December 6-7, 2017

Thematic Workshop

Computational challenges for mechanical simulations based on digital image data





Institute of Engineering Mechanics Chair for Continuum Mechanics

In collaboration with





EMMA – Efficient Methods for Mechanical Analysis

1 Scope

The simulation of microstructured solids in the context of mechanical multiscale problems remains a challenging field with many questions being unanswered. Particularly, the reduction of the computational cost (cpu time and memory) while limiting numerical artifacts induced by the employed discretization methods is still sought after. The thematic workshop aims at developing discussions and interactions within a familiar atmosphere. Therefore, a scientific program including not only presentations but also dedicated time slots for small group discussions is proposed.

1.1 Workshop topics

Workshop topics include but are not limited to

- FFT-based numerical methods
- microstructure generation
- microstructure characterization
- coupled problems

A special session entitled **Challenges and Perspectives** is included which is intended to initiate discussions about open questions and possible research directions.

1.2 Link to GAMM AG Data

One focus of the GAMM activity group Data-driven modeling and numerical simulation of microstructured materials is the simulation and characterization of materials based on digital image data. The thematic workshop intends to provide a forum for discussions focussed on this specific topic.

1.3 Workshop fees

In line with the scope of this workshop, we wish to keep the workshop as simple as possible (but not simpler). No workshop fees are raised, and the overhead is kept at a bare minimum. In particular, if you wish to join us for lunch at the KIT mensa, you will have to pay yourself.

Fortunately, we were able to fund a workshop dinner without any extravagances (including one meal and one beverage, further details below).

2 Organizers

Prof. Dr. Matti Schneider Assistant Professor for Computational Micromechanics Institute of Engineering Mechanics Karlsruhe Institute of Technology (KIT), Germany

Dr.-Ing. Dipl.-Math. techn. Felix Fritzen Head of Emmy-Noether Group EMMA – Efficient Methods for Mechanical Analysis University of Stuttgart, Germany

3 Contact

For any further questions, please contact: Helga Betsarkis Institute of Engineering Mechanics Kaiserstr. 10, Building 10.23 76131 Karlsruhe, Germany Tel.: +49 721 608-46107 Fax: +49 721 608-44187 E-Mail: helga.betsarkis@kit.edu URL: http://www.itm.kit.edu/cm

4 Support

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5 Timetable

Dec. 6	11:30-12:00	Informal meeting	KIT ITM
Dec. 6	12:00-13:00	Lunch	KIT mensa
Dec. 6	13:15-15:15	Session	"Tangentialraum"
		Innovative Simulation Methods	
Dec. 6	15:15-15:45	Coffee break	"Tangentialraum"
Dec. 6	15:45-17:45	Session	"Tangentialraum"
		Challenges and perspectives	
Dec. 6	18:30	Workshop Dinner	Hoepfner Burghof
Dec. 7	09:00-10:00	Session	"Tangentialraum"
		Microstructure characterization and	l generation
Dec. 7	10:00-10:30	Coffee break	"Tangentialraum"
Dec. 7	10:30-11:30	Session (ctd.)	"Tangentialraum"
		Microstructure characterization and	l generation
Dec. 7	11:30-12:00	Final discussion	"Tangentialraum"
Dec. 7	12:00-13:00	Lunch	KIT mensa

6 Locations

6.1 Getting to KIT Campus Süd

In front of the main exit of Karlsruhe Central Station, all trams passing by KIT leave in rightwards direction. Direct connections are offered by tram 2 ("Wolfartsweier") or S4 ("Bretten/Gölshausen" or "Heilbronn"). Get off at the tram stations "Kronenplatz" or "Durlacher Tor/KIT Campus-Süd".

6.2 KIT ITM

The Institute of Engineering Mechanics (ITM) is located on the third floor, building 10.23, see Fig. 1 or online.

6.3 KIT mensa

Lunch on Dec. 6 and Dec. 7 takes place at the KIT mensa, see Fig. 1. We have reserved a few tables in the large dining hall (at "Linie 5"), close to the windows. At "Linie 6", the "Update", you can pay cash. For the other options, you need a KIT student or employee card (or someone to carry you along).



Figure 1: Map of the KIT Campus Süd. The tram/train station, the mensa as well as the buildings 10.23 (KIT ITM) and 20.30 (Seminar room) are marked in blue.

6.4 Seminar room "Tangentialraum"

The workshop venue, the seminar room SR 1.067 ("Tangentialraum") is located on the first floor in the building 20.30.

6.5 Hoepfner Burghof

The conference dinner will take place at the "Hoepfner Burghof", Haid-und-Neu-Straße 18, 76131 Karlsruhe, starting from 18:30h on Dec. 6, 2017. According to google maps it takes about 20 minutes to reach the restaurant from the seminar room "Tangentialraum". Alternatively, you could use tram number 4 up to Karl-Wilhelm-Platz or Hauptfriedhof.

7 Schedule for the sessions

7.1 Session – Innovative Simulation Methods

This session takes place on Dec. 6, 2017, 13:15-15:15h with the following presentations:

13:15-13:45	F. S. Göküzüm, L. T. K. Nguyen and MA. Keip	
	Consistent FFT-based homogenization of	
	electromechanically coupled materials	
13:45-14:15	T. Hofmann, H. Andrä, R. Müller and J. Zausch	
	Stress simulation in lithium-ion batteries	
14:15-14:45	15-14:45 M. Osorno, M. Schirwon, N. Falkner,	
	R. Sivanesapillai, H. Steeb and D. Göddeke	
	Large scale SPH simulation of single phase flow through	
	porous media: Scalability study	
14:45-15:15	J. Kochmann	
	Two-scale FE-FFT-based finite strain crystal plasticity	
	simulations of polycrystalline materials	

7.2 Session – Challenges and perspectives

This session takes place on Dec. 6, 2017, 15:45-17:45h and contains three presentations. These are particularly intended to foster discussions.

F. Welschinger	Computational multi-scale methods for polymers in an industrial environment
V. Schmidt	Stochastic modeling of the 3D microstructure of hierarchical composite materials
F. Fritzen	How much data is actually needed?

7.3 Session – Microstructure characterization and generation

This session takes place on Dec. 7, 2017, 9:00-11:30h with the following presentations:

09:00-9:30	<u>M. Schneider</u> , T. Hofmann, H. Andrä, P. Lechner F. Ettemeyer, W. Volk and H. Steeb
	Modelling the microstructure and computing
	effective elastic properties of sand core materials
09:30-10:00	D. Westhoff
	Structural Segmentation of Defective Particle Systems
10:00-10:30	Coffee Break
10:30-11:00	M. Neumann
	On microstructure-property relationships derived by virtual materials
	testing with an application to Fickian diffusion in silica monoliths
11:00-12:00	Discussion and Closing

Consistent FFT-based homogenization of electromechanically coupled materials

Felix Selim Göküzüm¹, Lu Trong Khiem Nguyen¹ and Marc-André Keip¹

¹: Institute of Applied Mechanics (Civil Engineering), Chair I, University of Stuttgart, Stuttgart, Germany

Future challenges of engineering such as medical applications, robotics or space exploration demand for materials, which are tailor-made and adaptive. In this context, the group of electroactive polymers has recently attracted attention [1]. The behaviour of these materials is often implied by their specific microstructure. The prediction of the material's macroscopic behaviour based on microstructure information is of special interest for research and industry. In 1994 Moulinec & Suquet [2], suggested an FFT-based iterative scheme for predicting the mechanical response of heterogeneous materials. The present work extends the framework to electromechanically coupled materials, where the focus of the work lies on the formulation of a most general form of the preconditioned Lippmann-Schwinger equation. The role of the coupled reference medium on the convergence behaviour is investigated numerically. Additionally, we extend the consistent macroscopic tangent operator introduced in an earlier work to coupled problems. In this context, a fully coupled, a reduced non-coupled and a finite-difference based approach are compared to each other. The framework is formulated in a large-strain setting [3]. Finally, the model is tested in a FE-FFT multiscale simulation [4].

References

- [1] Y. Bar-Cohen: Electroactive polymers as artificial muscles: Capabilities, potentials and challenges, Robotics, 188–196 (2000).
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- [3] M. Kabel, T. Böhlke, M. Schneider: Efficient fixed point and Newton-Krylov solvers for FFT-based homogenization of elasticity at large strains, Computational Mechanics, 54, 1497–1514 (2014).
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Stress simulation in lithium-ion batteries

Tobias Hofmann¹, Heiko Andrä¹, Ralf Müller² and Jochen Zausch¹

- ¹: Fraunhofer Institute for Industrial Mathematics ITWM, Kaiserslautern, Germany
- ²: Chair of Applied Mechanics, TU Kaiserslautern, Germany

Phase separation during the intercalation of lithium ions can lead to degradation effects in some cathode materials. A model describing lithium ion diffusion, electric potentials and small deformations is introduced on the microscale. The Cahn-Hilliard equation is used in an electrochemical model coupled to linear elasticity of small strains in the electrode material. An immersed boundary method is used with adaptive time steps. The charging of porous microstructures is numerically simulated.

Large scale SPH simulation of single phase flow through porous media: Scalability study

<u>M. Osorno^{1,3}</u>, M. Schirwon^{2,3}, N. Falkner^{1,3}, R. Sivanesapillai^{1,3}, H. Steeb^{1,3} and D. Göddeke^{2,3}

¹: Institute of Applied Mechanics (CE), University of Stuttgart, Pfaffenwaldring 7, 70569 Stuttgart, Germany

²: Institute for Applied Analysis and Numerical Simulation, University of Stuttgart, Pfaffenwaldring 57, 70569 Stuttgart, Germany

³: Stuttgart Research Center for Simulation Technology, Pfaffenwaldring 5a, 70569 Stuttgart, Germany

Direct Numerical Simulation of fluid flow in high-resolution imaged geometries of real porous materials is highly relevant in several fields, such as, carbon dioxide sequestration, geothermal energy generation and groundwater contamination remediation.

One method commonly applied for DNS in these settings is the quasi-incompressible Smoothed Particle Hydrodynamics (SPH). SPH is a meshless Lagrangian method, highly suitable for modeling complex geometries and flow at high Reynolds numbers. However, SPH requires high computational resources, especially in the simulation of large domains.

To overcome these difficulties, we have implemented the SPH method in the highly optimized software HOOMD-blue, for both CPUs and GPUs. To overcome these difficulties, we have extended the highly optimized HOOMD-blue molecular dynamics software to support the SPH method. Our implementation targets both CPUs and GPUs. We present initial scalability studies of the SPH module for a variety of representative test cases, with the long-term goal to extend the simulations to large-scale high-resolution imaged porous media. The simulations are performed on the supercomputers Hazel Hen (HLRS – Germany) and Piz Daint (CSCS – Switzerland) for the CPU and GPU parallelization respectively.

Two-scale FE-FFT-based finite strain crystal plasticity simulations of polycrystalline materials

Julian Kochmann¹

¹: Institute of Applied Mechanics, RWTH Aachen, Germany

Recently two-scale FE-FFT-based methods have been proposed to predict the local and effective mechanical behavior of composite [1] and polycrystalline materials [2],[3] at small strains. More recently, Göküzüm and Keip [4] proposed a finite strain extension for composite materials. This work deals with finite strain crystal plasticity simulations of polycrystalline aggregates and the development of an efficient and numerically robust solution scheme for two-scale full-field problem settings.

References

- [1] J. Spahn, H. Andrä, M. Kabel, R. Müller: A multiscale approach for modeling progressive damage of composite materials using fast Fourier transforms, Computer Methods in Applied Mechanics and Engineering, 268, 871–883 (2014).
- [2] J. Kochmann, J. R. Mianroodi, S. Wulfinghoff, S. Reese, B. Svendsen; Two-scale FE-FFT and phase-field-based computational modeling of bulk microstructural evolution and macroscopic material behavior; Computer Methods in Applied Mechanics and Engineering; Elsevier; 2016; Vol. 305, 89-110
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Modelling the microstructure and computing effective elastic properties of sand core materials

<u>Matti Schneider</u>¹, Tobias Hofmann², Heiko Andrä², Philipp Lechner³, Florian Ettemeyer⁴, Wolfram Volk^{3,4} and Holger Steeb⁵

- ¹: Karlsruhe Institute of Technology, Institute of Engineering Mechanics
- ²: Fraunhofer ITWM, Kaiserslautern, Department of Flow and Materials simulation
- ³: Technical University of Munich, Institute of Metal Forming and Casting
- ⁴: Fraunhofer Research Institution for Casting, Composite and Processing Technology IGCV, Garching, Department of Casting Technology
- ⁵: Institute of Applied Mechanics (CE), University of Stuttgart

In this work we model sand core materials on the micro-meter scale, resolving individual sand grains and binding bridges, to obtain effective elastic moduli of the composite by computational homogenization, laying the foundations for investigating the strength properties of core blown parts with foundry applications.

We analyze sand core materials on the basis of X-ray micro-computed tomography (μ XRCT) images and extract a couple of sand grains from this volume image. These grains enter a packing algorithm which can generate granular packs with high packing fraction and incorporate sand grains with high complexity. Furnished with binder the resulting microstructures are investigated, deriving their effective elastic properties and studying the sensitivity w.r.t. the entering parameters. If a realistic range of elastic parameters of both sand grains and binder are plugged into the simulation, the agreement with experimentally obtained *P*-wave moduli is excellent.

Structural Segmentation of Defective Particle Systems

Daniel Westhoff¹

¹: Institute of Stochastics, Ulm University, Germany

A common approach for segmenting particle systems, as they are for example observed in lithium-ion battery electrodes, is using watershed techniques, see, e.g., [1]. Based on the corresponding distance map, particles are identified using a flooding starting from the local maxima. However, this approach does often not work for defective particle systems, i.e., if particles have cracks, holes or breakages. This is due to the fact that the defects influence the distance map and there is not exactly one local maximum for each particle anymore. Therefore, the algorithm developed in [2] is presented, which is able to first identify defects in the particle system and then segment the system into individual particles. For validation, we apply the algorithm to simulated data using a stochastic 3D model as described in [3]. By artificially inserting defects, we can quantify the accordance of the algorithm. Finally, the algorithm is applied to tomographic image data from a battery anode and cathode. In batteries, such defects can occur during thermal runaway, see, e.g., [4]. A segmentation of such a data set is possible using the presented algorithm, see Figure 2 for an example.



Figure 2: Segmentation of a particle system into individual particles (different colors) with defects highlighted using grey color.

References

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On microstructure-property relationships derived by virtual materials testing with an application to Fickian diffusion in silica monoliths

Matthias Neumann¹

¹: Institute of Stochastics, Ulm University, Germany

Based on virtual materials testing, which combines image analysis, stochastic microstructure modeling and numerical simulations, quantitative relationships between microstructure characteristics and effective conductivity can be derived. The idea of virtual materials testing is to generate a large variety of stochastically simulated microstructures in short time. These virtual, but realistic microstructures are used as input for numerical transport simulations. Finally, a large data basis is available to study microstructure-property relationships quantitatively by classical regression analysis and tools from statistical learning. The quantitative relationship,

$$\sigma_{\rm eff} = \sigma_0 \varepsilon^{1.15} \beta^{0.37} / \tau^{4.39} \tag{7.1}$$

between effective conductivity and the microstructure characteristics, which are relevant for transport (i.e. connected phase volume fraction ε , its mean geodesic tortuosity τ and constrictivity β) has been derived in [1]. Here σ_0 denotes the intrinsic conductivity of the material. An extension of this simulation study has been performed in [2] where 8119 virtual microstructures have been generated. Using methods from statistical learning as neural networks and random forest it can be shown that that it is possible to predict effective conductivity more accurately. Due to the mathematical analogy, the obtained results can be transferred from conduction processes to Fickian diffusion in order to predict the effective diffusivity in porous microstructures from silica monoliths [3], where effective diffusivity is simulated by a random-walk particle-tracking algorithm [4],[5]. In general, the proposed concept of virtual materials testing leads to a better understanding of microstructure-property relationships and thus, it has the potential to generate recommendations for new, more efficient microstructures.

References

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8 List of registered participants

name	institution	e-mail address
Heiko Andrä	Fraunhofer ITWM, KL	heiko.andrae@itwm.fraunhofer.de
Robert Bertoti	KIT	robert.bertoti@kit.edu
Elmira Birang	U Erlangen-Nuremberg	elmira.birang@fau.de
Thomas Böhlke	KIT	thomas.boehlke@kit.edu
Felix Dietrich	TU KL	fdietric@rhrk.uni-kl.de
Nadine Falkner	U Stuttgart	nadine.falkner@mechbau.uni-stuttgart.de
Felix Fritzen	U Stuttgart	felix.fritzen@mechbau.uni-stuttgart.de
Felix Göküzüm	U Stuttgart	felix.goekuezuem@mechbau.uni-stuttgart.de
Johannes Görthofer	KIT	johannes.goerthofer@kit.edu
Vinzenz Guski	U Stuttgart	vinzenz.guski@imwf.uni-stuttgart.de
Jörg Hohe	Fraunhofer IWM, FB	joerg.hohe@iwm.fraunhofer.de
Tobias Hofmann	Fraunhofer ITWM, KL	tobias.hofmann@itwm.fraunhofer.de
Loredana Kehrer	KIT	loredana.kehrer@kit.edu
Axel Klawonn	U Köln	axel.klawonn@uni-koeln.de
Jonathan Köbler	Fraunhofer ITWM, KL	jonathan.koebler@itwm.fraunhofer.de
Julian Kochmann	RWTH Aachen	julian.kochmann@rwth-aachen.de
Oliver Kunc	U Stuttgart	oliver.kunc@mechbau.uni-stuttgart.de
Matthias Leuschner	-	_
Dennis Merkert	TU KL	dmerkert@mathematik.uni-kl.de
Matthias Neumann	U Ulm	matthias.neumann@uni-ulm.de
Maria Osorno	U Stuttgart	maria.osorno@mechbau.uni-stuttgart.de
Oliver Rheinbach	TU Freiberg	oliver.rheinbach@math.tu-freiberg.de
Benedikt Prifling	U Ulm	benedikt.prifling@uni-ulm.de
Sahba Sadir	KIT	sahba.sadir@kit.edu
Frederik Scherff	U Saarland	frederik.scherff@uni-saarland.de
Volker Schmidt	U Ulm	volker.schmidt@uni-ulm.de
Matti Schneider	KIT	matti.schneider@kit.edu
Yanling Schneider	U Stuttgart	yanling.schneider@imwf.uni-stuttgart.de
Michael Schober	Fraunhofer IWM, FB	michael.schober@iwm.fraunhofer.de
Fabian Welschinger	Robert BOSCH GmbH	fabian.welschinger@de.bosch.com
Daniel Westhoff	U Ulm	daniel.westhoff@uni-ulm.de